# Infrared absorption spectra of evaporated V<sub>2</sub>O<sub>5</sub> and co-evaporated V<sub>2</sub>O<sub>5</sub>/B<sub>2</sub>O<sub>3</sub> thin films

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The Fourier transform infrared spectra of different compositions of evaporated  $V_2O_5/B_2O_3$  thin films have been investigated. Most of the absorption bands corresponding to  $V_2O_5$  and  $B_2O_3$ films coincide with those reported by other authors. The short-range order in amorphous  $V_2O_5$ films is found to be conserved. The absorption spectra indicate a boroxol ring structure **for**   $B_2O_3$  films. In co-evaporated  $V_2O_5/B_2O_3$  films the boron is observed to substitute in the  $V_2O_5$ network such that the coordination number of vanadium ion remains unchanged. The presence of a number of bands corresponding to  $-OH$  groups indicates the hygroscopic character of the films.

## **1. Introduction**

The molecule in a solid can undergo two different types of movements, rotational and vibrational. The total energy of a molecule in a solid in comprised of three different parts, the energy related to the rotational motion, that related to the vibrational motion, and the energy responsible for the motion of the electrons in the motecule itself. The energies corresponding to rotational spectra lie in the low frequency  $(1 \text{ to } 100 \text{ cm}^{-1})$  range because in this case the translations occur between the closely spaced rotational levels. On the other hand, the vibrational levels in solids are separated by comparatively greater energy differences, normally in the range (10<sup>4</sup> to  $10^2$  cm<sup>-1</sup>), so that the vibrational spectra of a solid can be recorded in the infrared range of energy.

Valuable information about the various energies taking part in the bonding can be achieved with the help of infrared absorption which is a useful measuring technique because it is one of the few methods that allow direct examination of the absorbed molecules on the solid surface. The infrared absorption spectra have been used to ascertain whether any kind of local disorder characteristic of the constituent oxides is maintained in the glass.

West [1] suggested that the  $B_2O_3$  glass contains  $BO<sub>3</sub>$  triangular units and borate glasses contain a mixture of  $BO_3$  triangles and  $BO_4$  tetrahedra, depending upon composition. According to him an important constituent of vitreous  $B_2O_3$  is the boroxol group which is a planar, six-membered ring of alternate boron and oxygen atoms. These groups are linked into a three-dimensional network by a bridging oxygen.

While studying the vibrational spectra of vitreous  $B_2O_3$ , Windisch and Risen [2] suggested that the boroxol rings involving three linked  $BO<sub>3</sub>$  triangular units are the major constituents of vitreous  $B_2O_3$ . Walrafen *et al.* [3] have shown that at higher temperatures, the networks of vitreous  $B_2O_3$  and of  $B_2O_3$ melts consist of  $BO<sub>3</sub>$  triangles as a basic structural unit

instead of boroxol rings. From the study of nuclear magnetic resonance (NMR) spectra, Jellison *et al.* [4] proposed a boroxol ring model for the structure of vitreous  $B_2O_3$ . Elliott [5] suggested a continuous random network (CRN) of planar  $BO<sub>3</sub>$  units in his model for vitreous  $B_2O_3$ . He disagreed with the proposed existence of the boroxol group in this material. Recently, Kanehisa and Elliott  $[6]$  proposed that  $B_2O_3$  glass is an infinite network of planar boroxol rings,  $B_3O_3$ , interconnected via oxygen bridges as proposed by West [1].

In  $V_2O_5$ , the  $V^{5+}$  ion exists in six-fold coordination. It is generally recognized that as long as the coordination of the cations remains the same, the infrared absorption bands of a given compound are the same [7, 8]. For a similar composition of  $V_2O_5$ , the absorption band corresponding to the V-O vibration will not change for both the crystalline and the vitreous state. If the absorption bands corresponding to crystalline and amorphous  $V_2O_5$  are similar, then it can be concluded that the  $V^{5+}$  ions in the glassy state of  $V_2O_5$  will also exist in six-fold coordination.

Rivoalen *et al.* [9] have suggested that the g-values in the electron spin resonance (ESR) spectra for both amorphous and orthorhombic  $V_2O_5$  are the same, indicating that the short-range order remains the same in both cases. They also proposed that the unpaired 3d electron is not always localized on a single vanadium nucleus. It moves from one nucleus to another according to the hopping process which is responsible for the semiconducting properties of the oxide. Kahn *et al.*  [10] reported a difference in the short-range order of crystalline and amorphous  $V_2O_5$  based on ESR investigations. On the other hand, and also by means of ESR studies, Henri *et al.* [11] reported that the shortrange order characteristic of crystalline  $V<sub>2</sub>O<sub>5</sub>$  is conserved in amorphous  $V_2O_5$ . From the infrared spectra of GeO<sub>2</sub>-P<sub>4</sub>O<sub>10</sub>-V<sub>2</sub>O<sub>5</sub> glasses, it is observed that  $V^{5+}$ ions do not change their coordination when the samples change from the crystalline state to the glassy

state [7]. Culea *et al.* [12] and Sharma *et al.* [13] investigated the infrared spectra of  $V_2O_5-B_2O_3$ glasses. We have studied the infrared spectra of both evaporated  $V_2O_5$  and co-evaporated  $V_2O_5/B_2O_3$  thin films,

### **2. Experimental procedure**

For infrared absorption studies, the evaporated films of  $V_2O_5$  and  $V_2O_5/B_2O_3$  were deposited on the substrates of silicon wafers using a co-evaporation tech- \_ nique as developed by Hogarth and Wright  $[14]$ . The Fourier transform infrared absorption spectra from  $400$  to  $4000 \text{ cm}^{-1}$  for evaporated films were recorded on a Perkin-Elmer 1710 Fourier transform infrared spectrometer. While recording the absorption spectra, an uncoated silicon wafer (substrate) was placed in the reference beam in order to compensate the effect of substrate. The spectrometer was set to record an absorption spectrum for each sample from an average of ten scans.

### **3. Results and discussion**

The Fourier transform infrared (FTIR) transmission spectra for evaporated  $V_2O_5$ ,  $B_2O_3$  and co-evaporated  $V_2O_5/B_2O_3$  films, each of thickness about 300 nm, are shown in Fig. 1. The spectrum of  $B_2O_3$ films was recorded immediately after the evaporation in order to minimize the hygroscopic effect. The



*Figure 1* FTIR spectra of vacuum-evaporated  $V_2O_5/B_2O_3$ (300 nm) thin films of different compositions: (a) simple  $V_2O_5$ , (b)  $80\%$ V<sub>2</sub>O<sub>5</sub>/20%B<sub>2</sub>O<sub>3</sub>, (c)  $70\%$ V<sub>2</sub>O<sub>5</sub>/30%B<sub>2</sub>O<sub>3</sub>, (d) 65%  $V_2O_5/35\%B_2O_3$ , (e) simple  $B_2O_3$ .

TABLE I Various absorption bands recorded in the Fourier transform infrared spectra of evaporated  $V_2O_5$  and co-evaporated  $V_2O_5/B_2O_3$  thin films; (sr) represents a shoulder and (sp) represents a sharp band-

100% $V_2O_5$ $\rm (cm^{-1})$	$80\% \text{ V}_2\text{O}_5$ $\rm (cm^{-1})$	$70\%$ $V_2O_5$ $\rm (cm^{-1})$	$60\% \text{ V}, \text{O},$ $\rm (cm^{-1})$	100% $B_2O_3$ $\rm (cm^{-1})$
460 (sr)	$480$ (sr)			470 (sr)
				$560$ (sp)
600 (sr)		$600$ (sr)	600(sr)	
660(sp)	660(sp)	660(sp)	$660$ (sp)	690 (sr)
	720(sp)	$720$ (sp)	.720 (sr)	$730$ (sp)
	900 (sr)	890(sr)	900 (sr)	
990 (sr)	990 (sr)	990(sp)	990(sp)	
	1100(r)	1120(sp)		1150 (sr)
	1290(sp)	1290 (sr)	1290 (sr)	1270(sp)
1440 <sub>(sp)</sub>		1440 (sr)	1420(sp)	1470(sp)
1620(sp)	$1630$ (sp)	$1630$ (sp)	1630(r)	

absorption bands corresponding to different compositions of  $V_2O_5/B_2O_3$  films are listed in Table I. Firstly we shall try to assign the different bands to the vibrations of various structural units of simple  $V_2O_5$  and  $B_2O_3$  films and then we shall consider the co-evaporated films. All compositions are quoted in mol%.

## 3.1.  $B_2O_3$  films

The various absorption peaks observed in the FTIR spectra of evaporated amorphous thin films of  $B_2O_3$ can be seen in Table I. While investigating the infrared spectra of  $V_2O_5-B_2O_3$  glasses, Sharma *et al.* [13] recorded the different bands for the crystalline  $B_2O_3$ at 545, 650, 740 to 800, 890, 960, 1020, 1195, 1430 to 1500, 2260, 2360, 2560 and 3200 cm -1. On the other hand, during the study of infrared spectra of  $V_2O_5-B_2O_3$  glasses, Culea *et al.* [12] recorded the bands at 565, 800, 1200, 1420, 1640, 3215 and 3405 cm<sup>-1</sup> for vitreous  $B_2O_3$ .

We observe that in our samples of evaporated thin films of  $B_2O_3$  films, most of the bands recorded by other authors [12, 13] are present, although there are some generally small shifts in the frequencies corresponding to some of the bands. The different bands observed by us for amorphous films of  $B_2O_3$  may be related to different modes of vibrations of structural units as below  $\lceil 15, 16 \rceil$ .

The absorption band around  $470 \text{ cm}^{-1}$  may be related to the ring-angle bending [17, 18]. The band at 560 cm<sup>-1</sup> is related to the vibration of the B-O-B bending motion [17]. The shoulder corresponding to about 690 cm<sup> $-1$ </sup> can be assigned to B-O' wagging [18] where O' does not belong to a ring.

The sharp band appearing at  $730 \text{ cm}^{-1}$  arises from the bond bending motion of B-O-B centres in the network [17, 19, 20]. The absorption band appearing at 1260 cm<sup>-1</sup> can be attributed to the symmetric B-O' stretching vibrations [18]. Tenney and Wong also found this strong band in  $B_2O_3$  [19]. The bands at 1150 and  $1470 \text{ cm}^{-1}$  may be related to B-O-H bending and B-O stretching vibrations, respectively, as proposed by Parsons and Milberg [17].

The bands observed by us at the higher wavenumbers for evaporated simple  $B_2O_3$  and  $V_2O_5$  and co-evaporated  $V_2O_5/B_2O_3$  films (not shown in the Table I) are evidently summation or overtone bands involving B-O and V-O bending and stretching vibrations. Such bands have also been observed by Parsons and Milberg [17] and Krogh-Moe [18] for  $B_2O_3$ glasses. A very sharp and strong band near  $3200 \text{ cm}^{-1}$ indicates the presence of  $-OH$  groups which may be present due to the hygroscopic character of  $B_2O_3$ films. Tomozawa and Tomozawa [21] suggested that the peak at 3600 cm<sup> $-1$ </sup> may be related to free hydroxol while that at  $2750 \text{ cm}^{-1}$  is due to hydrogen-bonded hydroxol in borosilicate glasses.

With the support of these observations, we propose the structure of our evaporated  $B_2O_3$  film to consist of boroxol rings linked by bridging oxygen as shown below.



Our proposed structure agrees with that suggested by West [1] and Kanehisa and Elliott [6].

## 3.2. V<sub>2</sub>O<sub>5</sub> films

The first column of Table I shows the absorption bands in the FTIR spectra of evaporated amorphous thin films of  $V_2O_5$ . Sharma *et al.* [13] recorded the infrared absorption bands at 300, 370, 470, 590, 825 and 1025 cm<sup>-1</sup> for crystalline  $V_2O_5$ . They also noted that water bands or -OH stretching modes were not present in the infrared spectrum of crystalline  $V_2O_5$ . Culea *et al.* [12] recorded two main infrared absorption bands at  $825$  and  $1020 \text{ cm}^{-1}$  for crystalline  $V<sub>2</sub>O<sub>5</sub>$ . Hogarth and Hosseini [22] observed absorption peaks in the infrared spectra of crystalline  $V_2O_5$ at 280, 370, 480, 660, 800, 1040, 1260, and 1275 cm<sup>-1</sup>. They related the relatively sharp peak at  $1040 \text{ cm}^{-1}$  to the V-O stretching frequency and further suggested that the V-O stretching frequency for the compounds in which the vanadium atoms are totally ionized to  $V^{5+}$  lies in the range 1025 to 1005 cm<sup>-1</sup>, and for  $V^{4+}$ lies in the range 900 to 1025 cm<sup> $-1$ </sup>. Two narrow bands at the frequencies 1040 and 985 cm<sup> $-1$ </sup> were recorded by Hevesi *et al.* [23] in the infrared spectra of  $V_2O_5$ single crystals. They attributed these bands to the transition of vanadium ions from ground level to an excited level.

It has been observed that the band corresponding to the stretching vibration of a V-O bond appears at 1015 cm<sup>-1</sup>. The presence of  $V^{4+}$  ions appears to reduce the V-O stretching frequency and it lies between 1015 and 900 cm<sup>-1</sup> [24]. We propose that in our case of evaporated thin films of  $V_2O_5$ , one shoulder which appears at 990 cm<sup>-1</sup> is due to the V-O stretching

vibration mode  $[7, 25-27]$ . The presence of the peak at about 990  $\text{cm}^{-1}$  in all our samples indicates that the site symmetry of the vanadium cation in glasses may be similar to that in crystalline  $V_2O_5$ .

It is reported by Dimitriev *et al.* [26] that vitreous  $V_2O_5$  can contain more than one short V=O bond, having two close absorption frequencies. The bands at 660 and 600 cm<sup>-1</sup> may be attributed to V=O double bonds as suggested by Hogarth and Ahmad [28]. This is also in agreement with the suggestion made by Dimitriev *et al.* [26]. The bands appearing at frequencies around and higher than  $1440 \text{ cm}^{-1}$  may be attributed to the summation and overtone bands corresponding to stretching vibrations of V-O and V=O bonds. These types of band have also been recorded by Sharma *et al.* [13] for  $V_2O_5 - B_2O_3$  glasses.

## 3.3.  $V_2O_5/B_2O_3$  films

Columns 2, 3 and 4 of Table I show the various absorption bands for different compositions of coevaporated  $V_2O_5/B_2O_3$  thin films. Almost all the main absorption peaks observed for the spectrum of evaporated  $B_2O_3$  films are also observed to be present in the spectra of co-evaporated  $V_2O_5/B_2O_3$  thin films.

While studying the infrared spectra of lithium borosilicate glasses with  $V_2O_5$ , Sanad *et al.* [29] recorded bands at 1375, 1080, 940, 890, 790, and  $760 \text{ cm}^{-1}$ . They related the band at 980 cm<sup>-1</sup> to the presence of non-bridging oxygen vibrations and the bands at 790 and 1220 cm<sup>-1</sup> to the stretching vibration of the boroxal ring. The strong and wide bands around 1420 and 3200 cm<sup> $-1$ </sup> which appear in composite  $V_2O_5/B_2O_3$  films such as  $V_2O_5-B_2O_3$  glasses, may be attributed to the bending and stretching vibration of the OH group [12] which may be present in our samples because of their hygroscopic nature.

The shoulder observed at  $990 \text{ cm}^{-1}$  for simple  $V<sub>2</sub>O<sub>5</sub>$  films is converted into a sharp band on increasing the content of  $B_2O_3$  in co-evaporated  $V_2O_5/B_2O_3$ films as can be seen in Fig. 1. This band can be attributed to V-O stretching as observed by Anderson and Compton [25]. Dimitriev *et al.* [26] suggested that in  $V_2O_5-B_2O_3$  glasses, boron participates only in the substitutional positions without interacting with an isolated V=O bond and because of this reason no changes occur in the high ware number of infrared spectra. In the present case, the position of this band remains almost unchanged for all compositions of coevaporated  $V_2O_5/B_2O_3$  films and that for simple  $V_2O_5$  which indicates that the co-ordination of vanadium remains unchanged. We propose that the boron ions substitute in the network in the manner as shown below.



A similar structure has also been suggested by Dimitriev *et al.* [26].

The new peak at  $890 \text{ cm}^{-1}$  which appears on adding  $B_2O_3$  to  $V_2O_5$  in co-evaporated  $V_2O_5/B_2O_3$  and is present in all the co-evaporated samples is attributed to the boroxol ring structures. It also indicates that the boroxol ring is conserved in all co-evaporated  $V_2O_5/B_2O_3$  films.

The band appearing at  $1620 \text{ cm}^{-1}$  is present in all compositions of co-evaporated  $V_2O_5/B_2O_3$  films. This band shifted slightly towards the higher frequencies on increasing the  $B_2O_3$  content and may also be attributed to -OH bending vibrations due to their hygroscopic nature. A similar type of band has also been recorded by Sharma *et al.* [13] for  $V_2O_5 - B_2O_3$ glasses. This band was absent in the spectra of  $V_2O_5-B_2O_3$  glasses as recorded by Culea *et al.*  $[12]$ .

For simple  $V_2O_5$  there is a weak shoulder around 800 cm<sup>-1</sup>. This band is absent for  $V_2O_5$  contents up to 80% in co-evaporated  $V_2O_5/B_2O_3$  films. On lowering the  $V_2O_5$  content further, this band reappears and becomes sharper on increasing the  $V_2O_5$ content.

Sharma *et al.* [13] reported a broad absorption peak between 740 and 800 cm<sup>-1</sup> in crystalline  $B_2O_3$ and at 825 cm<sup>-1</sup> for crystalline  $V_2O_5$ . It has been reported [9] that there is no apparent change in the frequencies corresponding to the bands for crystalline and amorphous  $V_2O_5$ . In the present studies, for coevaporated  $V_2O_5/B_2O_3$  films the main absorption bands corresponding to simple  $B_2O_3$  and  $V_2O_5$ simple films are conserved, although their intensity goes on decreasing on increasing the  $B_2O_3$  content in the co-evaporated films. From the appearance of a band around 900 cm<sup> $-1$ </sup>, we conclude that the structural unit of  $V_2O_5$  becomes more infrared active on increasing the content of  $B_2O_3$  in co-evaporated  $V_2O_5/B_2O_3$  films.

A number of small bands appearing at 490, 560 cm<sup>-1</sup> for simple  $B_2O_3$  films are absent in the coevaporated  $V_2O_5/B_2O_3$  films suggesting some sort of structural re-arrangements of  $B_2O_3$  during the coevaporation process. We propose that the band at 690 cm<sup>-1</sup> recorded for simple  $B_2O_3$  films may be conserved in co-evaporated  $V_2O_5/B_2O_3$  films, but it is indistinguishable from that at  $660 \text{ cm}^{-1}$  corresponding to V=O stretching frequency.

The sharp band appearing at  $730 \text{ cm}^{-1}$  for simple B203 films becomes a wider band and moves towards the lower frequency for co-evaporated  $V_2O_5/B_2O_3$ films, indicating some sort of bonding changes in boron and oxygen-deformed sublattices.

The band at 1270 cm<sup>-1</sup> for  $B_2O_3$  films also becomes wider and moves towards the higher frequencies on increasing the  $B_2O_3$  content in co-eyaporated  $V_2O_5/B_2O_3$  films. The new band appearing at 1100 to 1120 cm<sup>-1</sup> for co-evaporated  $V_2O_5/B_2O_3$  films may be attributed to the loss of oxygen by  $V_2O_5$  to convert itself into  $V_2O_4$ . This spare oxygen may be used to yield BO<sub>4</sub> tetrahedra in  $V_2O_5/B_2O_3$  films. A similar type of band has also been observed by Quan and Adams [30] and Borrelli *et al.* [31] for rubidium borate and  $Na<sub>2</sub>O-B<sub>2</sub>O<sub>3</sub>$  glasses, respectively.

#### **4. Conclusion**

The evaporated thin films of simple  $B_2O_3$  and simple  $V<sub>2</sub>O<sub>5</sub>$  show infrared absorption bands which have been commonly observed during previous studies for these materials. For co-evaporated  $V_2O_5/B_2O_3$  films the main absorption bands corresponding to  $B_2O_3$ and  $V_2O_5$  are present with a small frequency shift and decreased intensity. Boron is substituted in the network such that the co-ordination number of vanadium is not changed. The  $B_2O_3$  films show a boroxol ring type structure.

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#### **References**

- 1. A.R. WEST, "Solid State Chemistry and its Applications" (Wiley, New York) p. 609.
- 2. C. F. WlNDISCH and W. M. RISEN, *J. Non-Crystalline Solids* 48 (1982) 307.
- 3. G. E. WALRAFEN, M. S. HOKMABADI, P. N. KRISHNAN and S. GUHA, *J. Chem. Phys.* 79 (1983) 3609.
- 4. G.E. JELLISON, L. W. PANEK, P. J. BRAY and G. B. ROUSE, *ibid.* 66 (1977) 802.
- 5. S.R. ELLIOTT, *Phil. Mag. B* 37 (1978) 435.
- 6. M. A. KANEHISA and R. J. ELLIOTT, Mater. Sci. Engng B3 (1989) 163.
- 7. B. H. V. JANAKIRAMA-RAO, *J. Amer. Ceram. Soc.* 49 (1966) 605.
- 8. F. DACHILLE and R. ROY, *ibid.* 42 (1959) 78.
- 9. L. RIVOALEN, A. REVCOLEVSCHI, J. LIVAGE and R. COLLONGUES, *J. Non-Crystalline Solids* 21 (1976) 171.
- 10. A. KAHN, J. LIVAGE and R. COLLONGUES, *Phys. Status Solidi (a)* 26 (1974) 175.
- 11. *M. HENRI, C.R'KHAandJ. LIVAGE, J. Phys. C14(1981)*  829.
- 12. E. CULEA, AL. NICULA and I. BRATU, *Phys. Status Solidi*  (a) 83 (1984) K15.
- 13. B.K. SHARMA, D. C. DUBE and A. MANSINGH, *J. Non-Crystalline Solids* 65 (1984) 39.
- 14. C.A. HOGARTH and L. A. WRIGHT, in "Proceedings of the 9th Conference on Physics Semiconductors", Moscow (Nauka, Leningrad, 1968) p. 1274.
- 15. G.C. DUBEY and M. GOPAL RAO, *Thin Solid Films* 79 (1981) L77.
- 16. A.P. IRWlN, J. S. HOLMGREN, T. W. ZERDA and J. JONAS, *J. Non-Crystalline Solids* 89 (1987) 191.
- 17. J.L. PARSONS and M. E. MILBERG, *J. Amer. Ceram. Soc.*  43 (1960) 326.
- 18. J. KROGH-MOE, *Phys. Chem. Glasses* 6 (1965) 46.
- 19. A.S. TENNEY and J. WONG, *J. Chem. Phys.* 56 (1972) 5516. 20. M. NOGAMI and Y. MORIYA, *J. Non-Crystalline Solids* 48
- (1982) 359.
- 21. H. TOMOZAWA and M. TOMOZAWA, *ibid.* 109 (1989) 311.
- 22. C.A. HOGARTH and A. A. HOSSEINI, *J. Mater. Sci. Lett. 3*  (1984) 359.
- 23. I. HEVESI, A. A. ABDULLAEV and V. S. LEBEDEV, *Sov. Phys. Crystallogr.* 16 (1971) 275.
- 24. M.N. KHAN, RAVISHANKAR HARANI, M. M. AHMAD and C. A. HOGARTH, *J. Mater. Sci.* 20 (1985) 2207.
- 25. G.W. ANDERSON and W. D. COMPTON, *J. Chem. Phys.*  52 (1970) 6166.
- 26. Y. DIMITRIEV, V. DIMITROV, M. ARNAUDOV and D. TOPLALOV, *J. Non-Crystalline Solids* 57 (1983) 147.
- 27. Y. IVANOVA and V. DIMITROV, *J. Mater. Sci. Lett. 2*  (1983) 541.
- 28. C.A. HOGARTH and M. M. AHMAD, *J. Mater. Sci. Lett. 2*  (1983) 649.
- 29. A.M. SANAD, F. A. MOUSTAFA, A. A. EL-SHARKAWY,

A. G. MOSTAFA, A. A. EL-SAGHIER and I. KASHIF, *J. Mater. Sci.* 23 (1988) 1553.

- 30. J. T. QUAN and C. E. ADAMS, *J. Phys. Chem.* 70 (1966) 331.
- 31. N. F. BORRELLI, B. D. McSWAIN and GOUQ-JEN SU, *Phys. Chem. Glasses 4* (1963) 11.

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